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STRUCTURE DETERMINATION OF CROSSPOLYMERIZED

POLY (1, 11 - DODECADIYNE),

12) 22/ (11) 3 May 12

Jerome B./Lando and M./Thakur Department of Macromolecular Science Case Western Reserve University Cleveland, Ohio 44106

May 5, 1981

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Electron diffraction patterns were obtained from crosspolymerized crystals of poly (1,11-dodecadiyne), formed by casting thin (>200Å) films of uncrosspolymerized polymer from chloroform. Two orientations of these crystals were obtained by varying the evaporation rate of the chloroform solvent. Crosspolymerization resulted from subsequent exposure of these samples to Coradiation. Thirty-six independent reflections were obtained from the a*c*					
reciprocal lattice net and a lattice net having b* and the (h 0 h) reciprocal					

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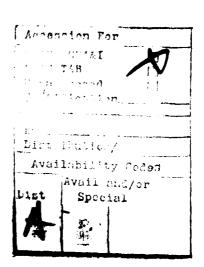
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lattice direction as orthogonal axes. Refinement of the structure was accomplished with these data. The unit cell of the crosspolymerized material was monoclinic, space group P2/n, a = 9.17A, b (hydrocarbon chain axis and unique axis) = 12.15 Å, c (polydiacetylene axis) = 9.92Å and the angle  $\beta = 123.5^{\circ}$ .

Late

## INTRODUCTION

The crosspolymerization of the macromonomer poly (1, 11 - dodecadiyne), using UV, x-ray or  ${\rm Co}^{60}$   $\gamma$ -radiation has recently been reported (1). The term macromonomer is used to describe the original polymer which has a chemical repeat unit  $\{({\rm CH_2})_8\text{-C=C-C=C}\}$ . The term crosspolymerization is utilized to distinguish systematic polymerization of the diacedylene units to a crystalline structure composed of sheets, as indicated in Fig. 1, from the more familiar random crosslinking that many polymers undergo when exposed to radiation. The crystal structure of the crosspolymerized material was refined using electron diffraction data because of the limited information obtained from x-ray fiber patterns.



### EXPERIMENTAL SECTION:

The macromonomer was prepared by oxidative coupling of HCΞC - (CH<sub>2</sub>)<sub>8</sub> - CΞCH using copper-pyridiene catalyst.<sup>1,2,3</sup> After purification<sup>1</sup> the polymer was dissolved in choloroform to make a dilute solution. A drop of this solution at 4°C, on a carbon coated copper grid, was evaporated to obtain single crystals. These crystals were then irradiated by γ-radiation (100 M rad) for complete crosspolymerization. Diffraction patterns were obtained by a JEOL JEM 100 B electron microscope at 100 KV. A second orientation of the crystals was obtained by casting a film at room temperature. Electron diffraction data were obtained at low beam intensity. The intensity data were collected using a high precision photodensitometer. The peaks in the intensity spectra were quite sharp; therefore the peak heights were taken as the relative intensities of the diffraction maxima. The structure was refined using the 'Lals Mark Six' program originally developed by Arnott and co-workers.<sup>4</sup>

Measurements of the thickness of the crystals was performed in the following way. The crystals on the carbon coated copper grid were carbonshadowed at an angle of 45°. Some part of the substrate remains unexposed to this shadowing due to the height of the crystal which blocks the carbon particles. The unexposed strip manifests itself as a light strip at the edge of the crystal on the electron micrograph. Since the shadowing angle is 45° the width of this strip is the thickness of the crystal.

### RESULTS AND DISCUSSION:

The diffraction pattern of the a\*c\* reciprocal lattice net remained sharp for about 20 secs but that of  $b*(h0\bar{h})$  net (rapid evaporation) remained reasonably sharp for only about 13 secs even under the lowest possible beam-intensity. In spite of this we obtained a reasonable number of reflections by developing the pictures appropriately. Initially only the a\*c\* data were used for refinement of the structure. Afterwards the full data were utilized. The equatorial hon reflections, which appeared on both patterns, were used for scaling. The a\*c\* pattern showed a c-axis repeat distance of 9.92Å which is approximately twice the ideal polydiacetylene chain repeat. Therefore one can safely assume that the diacetylene chain is along the c-axis. The (h0h) pattern as well as x-ray studies gave a b-axis of 12.25Å which obviously fits with the hydrocarbon chain repeat. This axis is orthogonal to the ac plane indicating possible chain folding in the macromonomer. Therefore the unit cell is monoclinic. There is a screw axis along b as demonstrated by the absence of 0k0 k = odd reflections. The  $h0\ell$  plane contains additional systematic absences, h+ $\ell$  = odd, which indicates a glide plane perpendicular to the b-axis. With this information the space group assigned to this crystal is P2,/n.

With the above information at hand we can conveniently choose our starting model as in Fig. 3. We place the diacelylene chain along the c-axis and the hydrocarbon chain along the b-axis. There are four repeat units per unit cell and an n-glide in the ac plane.

Therefore there should be one chain at the center of the unit cell.

A series of models, which were small modifications of our preliminary model, were tried. Initially all these models were tested against contact calculations (van der Waals interaction). All possible combinations of the four chains with regard to their senses (UP or DOWN) were taken into consideration. Orientation of the diacelylene rod with respect to the c-c planer zigzag was varied until contacts were minimized. Both carbon and hydrogen atoms were used for structure factor calculation. Since the measured thickness of the crystals is less than 200Å, dynamical effects were ignored. Or to be more precise, the contribution of the dynamical effect was considered to be within experimental error. The existence of systematic absenses was a strong indication that this was a reasonable assumption.

After expoliting the appropriate options provided by Lals we reached a residual of 0.13. The residual we refined is defined as

$$R = \left\{ \sum_{m}^{N} w_m \Delta F_m^2 / \sum_{m}^{N} w_m F_m^2 (obs) \right\}^{1/2}$$

$$m=1 \qquad m=1$$

where 
$$\Delta F_m^2 = |F_m^2(obs) - F_m^2(cal c)|$$
.

A comparison of calculated and observed structure factors is given in Table 1.

We kept  $W_m$  = 1 for all diffraction maxima. The projection on the ac plane (Fig. 4) shows that the diacetylene chains remain in the plane of c-c zigzag, which should be energetically favorable. Two UP and two DOWN chains was found to yield the best residual. A DOWN chain is formed by rotating an UP chain by 180° about the a-axis. The ac projection of the DOWN chain is almost identical with that of the UP chain, although

the two chains are not identical (bc projection, Figure 5). This similarity might mislead one to think that the unit cell should be half of what is given.

Final bond angles, dihedral angles, and the bond lengths of the single, double and triple bonds are given in Table 2. The labels of atoms in Table 2 are identical to those in Figure 5.

Single crystal electron diffraction patterns of the predominantly uncrosslinked macromonomer were obtained using high speed x-ray film under very low beam intensity. The unit cell (Table 3) is monoclinic and larger than that of the crosspolymerized polymer. Macroscopically this kind of contraction was observed when the unreacted macromonomer was crosspolymerized. This kind of behaviour has been observed for some other diacetylene polymerizations as well. The space group symmetry is unaltered by crosspolymerization. Thus the starting model for refining the macromonomer should not be much different from that of the crosspolymerized one. This work is now under way.

### ACKNOWLEDGEMENT:

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- 5. Day, David R.; Lando, J. B., Macromolecules, 13, 1483 (1980).

### LEGEND

- 1) Model of the Reaction of the Macromonomer.
  - A. Macromonomer
  - B. Crosspolymerized
- 2) Electron Diffraction Pattern (a\*c\* reciprocal lattice net).
- 3) Crosspolymerized Poly (1. 11 Dodecadiyne), Model Structure
- 4) ac Projection of Final Structure, t, triple bond.
- 5) bc Projection of Final Structure. t, triple bond, d, double bond.

TABLE 1

2       0       0       8099       7386         4       0       0       2369       2161         0       2       0       1532       1587         0       4       0       1202       1487         0       6       0       865       1143         0       8       0       682       887         0       0       2       5109       5188         0       0       4       817       1062         1       0       -1       3031       3357         1       0       1       2575       2734         -1       0       3       3458       3479         -1       0       3       3458       3479         -1       0       5       526       671         1       0       3       2357       2614         3       0       1       1052       1318         2       0       2       2345       2136         2       0       2       2345       2136         2       0       2       2345       2136         1       4       -1<	h	k	1	F - CALC	F - OBS
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0       6       0       865       1143         0       8       0       682       887         0       0       2       5109       5188         0       0       4       817       1062         1       0       -1       3031       3357         1       0       1       2575       2734         -1       0       3       3458       3479         -1       0       5       526       671         1       0       3       2357       2614         3       0       1       1052       1318         2       0       2       2345       2136         2       0       -2       5329       5140         -2       0       4       949       1294         1       2       -1       572       763         1       4       -1       493       366         1       4       -1       440       898         3       1       -3       271       234         2       1       -2       814       244         -1       2       1	0	2	0	1532	1587
0       8       0       682       887         0       0       2       5109       5188         0       0       4       817       1062         1       0       -1       3031       3357         1       0       1       2575       2734         -1       0       3       3458       3479         -1       0       5       526       671         1       0       3       2357       2614         3       0       1       1052       1318         2       0       2       2345       2136         2       0       2       2345       2136         2       0       -2       5329       5140         -2       0       4       949       1294         1       2       -1       572       763         1       4       -1       493       366         1       4       -1       249       122         2       1       -2       814       244         -1       2       1       293       548         2       2       2	0	4	0	1202	1487
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0       0       4       817       1062         1       0       -1       3031       3357         1       0       1       2575       2734         -1       0       3       3458       3479         -1       0       5       526       671         1       0       3       2357       2614         3       0       1       1052       1318         2       0       2       2345       2136         2       0       2       2345       2136         2       0       -2       5329       5140         -2       0       4       949       1294         1       2       -1       572       763         1       4       -1       493       366         1       4       -1       440       898         3       1       -3       271       234         1       1       -1       249       122         2       1       -2       814       244         -1       2       1       293       548         2       2       2	0	8	0	682	887
1       0       -1       3031       3357         1       0       1       2575       2734         -1       0       3       3458       3479         -1       0       5       526       671         1       0       3       2357       2614         3       0       1       1052       1318         2       0       2       2345       2136         2       0       -2       5329       5140         -2       0       4       949       1294         1       2       -1       572       763         1       4       -1       493       366         1       4       -1       440       898         3       1       -3       271       234         1       1       -1       249       122         2       1       -2       814       244         -1       2       1       293       548         2       2       -2       539       601         -2       2       2       1063       1221         3       2       -3 <td>0</td> <td>0</td> <td>2</td> <td>5109</td> <td>5188</td>	0	0	2	5109	5188
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-1       0       3       3458       3479         -1       0       5       526       671         1       0       3       2357       2614         3       0       1       1052       1318         2       0       2       2345       2136         2       0       -2       5329       5140         -2       0       4       949       1294         1       2       -1       572       763         1       4       -1       493       366         1       6       -1       440       898         3       1       -3       271       234         1       1       -1       249       122         2       1       -2       814       244         -1       2       1       293       548         2       2       -2       539       601         -2       2       2       1063       1221         3       2       -3       976       856         4       2       -4       257       671         4       0       -4	1	0	-1	3031	3357
-1       0       5       526       671         1       0       3       2357       2614         3       0       1       1052       1318         2       0       2       2345       2136         2       0       -2       5329       5140         -2       0       4       949       1294         1       2       -1       572       763         1       4       -1       493       366         1       6       -1       440       898         3       1       -3       271       234         1       1       -1       249       122         2       1       -2       814       244         -1       2       1       293       548         2       2       -2       539       601         -2       2       2       1063       1221         3       2       -3       976       856         4       2       -4       257       671         4       0       -2       2431       2734         4       0       -4	1	0	1	2575	2734
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3       2       -3       976       856         4       2       -4       257       671         4       0       -2       2431       2734         4       0       -4       1091       1245         3       0       -3       2864       3113         5       0       -5       1554       1659	2	2	-2	539	601
4     2     -4     257     671       4     0     -2     2431     2734       4     0     -4     1091     1245       3     0     -3     2864     3113       5     0     -5     1554     1659	-2	2	2	1063	1221
4       0       -2       2431       2734         4       0       -4       1091       1245         3       0       -3       2864       3113         5       0       -5       1554       1659	3	2	-3	976	856
4     0     -4     1091     1245       3     0     -3     2864     3113       5     0     -5     1554     1659	4	2	-4	257	671
3     0     -3     2864     3113       5     0     -5     1554     1659	4	0	-2	2431	2734
5 0 -5 1554 1659	4	0	-4	1091	1245
	3	0	-3	2864	3113
5 0 -3 1293 1525	5	0	<b>-</b> 5	1554	1659
	5	0	-3	1293	1525

Table 2 continued

h	k	1	F - CALC	F - OBS
-3	0	5	943	1318
3	0	-1	1430	1807
6	0	-4	711	793

TABLE 2

Atoms Constructing the angle	Bond Angle	Dihedral Angle
$c^0 c^1 c^2$	110.5	
$c^1 c^2 c^3$	109.6	179.3
$c^2 c^3 c^4$	108.8	180.0
$c^3 c^4 c^5$	110.5	180.0
$c^4 c^5 c^6$	112.3	180.0
$c^5 c^6 c^7$	112.5	180.0
$c^6 c^7 c^8$	112.5	180.0
$c^7 c^8 c^9$	108.5	180.0
$c^8 c^9 c^{10}$	112.0	180.0
$c^3 c^4 c^{11}$	124.0	356.1
$c^4 c^{11} c^{12}$	180.0	358.2
$c^4 c^5 H^{5A}$	110.0	60.0
$c^4 c^5 H^{5B}$	112.1	302.5
с <sup>5</sup> с <sup>6</sup> н <sup>6A</sup>	107.4	60.9
с <sup>5</sup> с <sup>6</sup> н <sup>6</sup>	106.6	300.0
$C^6 C^7 H^{7A}$	105.9	60.0
$c^6 c^7 H^{7B}$	106.4	300.0
$c^7 c^8 H^{8A}$	113.6	60.0
$c^7 c^8 H^{8B}$	113.2	301.2
с <sup>8</sup> с <sup>9</sup> н <sup>9А</sup>	106.1	61.2
ა <sup>3</sup> с <sup>9</sup> н <sup>9В</sup>	106.5	300.0
c <sup>9</sup> c <sup>10</sup> H <sup>10A</sup>	104.5	60.0
с <sup>9</sup> с <sup>10</sup> н <sup>10В</sup>	106.8	300.0
$c^3 c^2 H^{2A}$	105.3	61.4
$c^3 c^2 H^{2B}$	107.9	298.5
$c^2 c^1 H^{1A}$	110.8	60.0
$c^2 c^1 H^{1B}$	111.2	300.0

Table 2 continued

Atoms Constructing the angle	Bond Angle	Dihedr	Dihedral Angle		
$C^1 C^0 H^{0A}$	106.4	60.0			
$c^1 c^0 H^{OB}$	107.6	300.0			
	C-C	C = C	C≅C	С - Н	
Bond length $(\stackrel{Q}{A})$	1.534	1.33	1.20	1.065	
(Not refined)					

TABLE 3

Cell Parameters	Macromonomer (unreacted)	Crosspolymerized Material
а	13.25A	9.17A
ь	14.15Å	12.25Å
c	7.63Å	9.92Å
β	118.5°	123.5°

Unit cell data

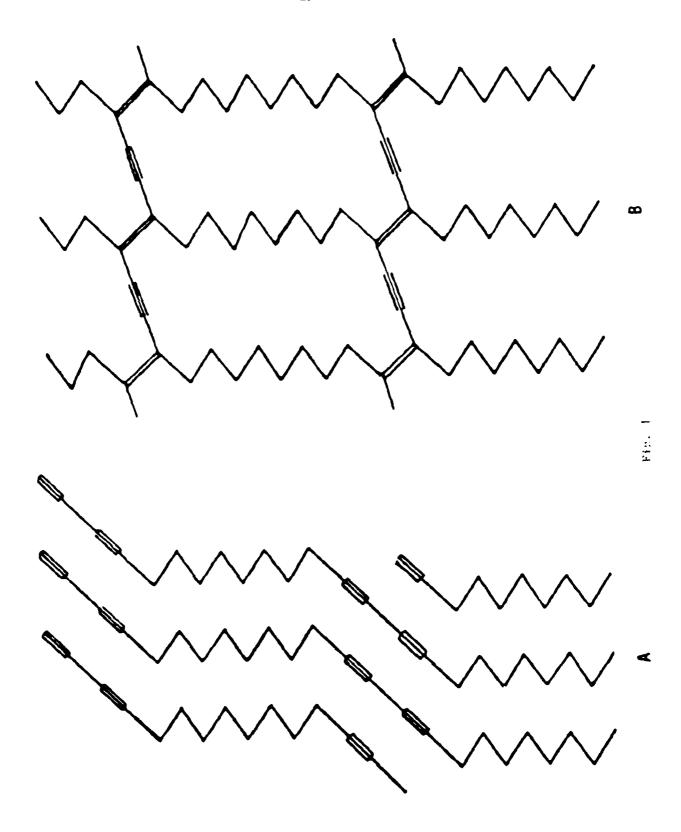




Fig. 1

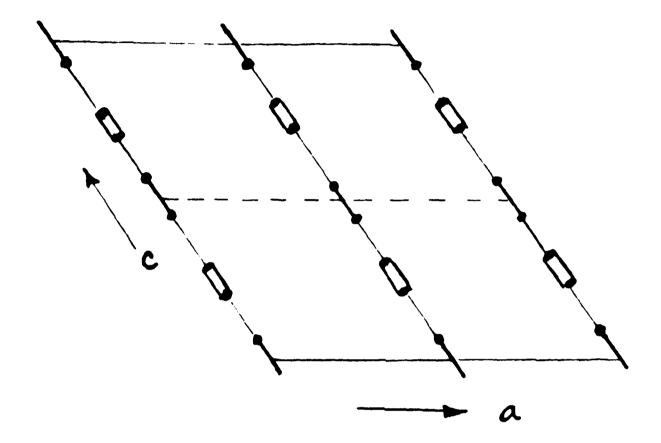
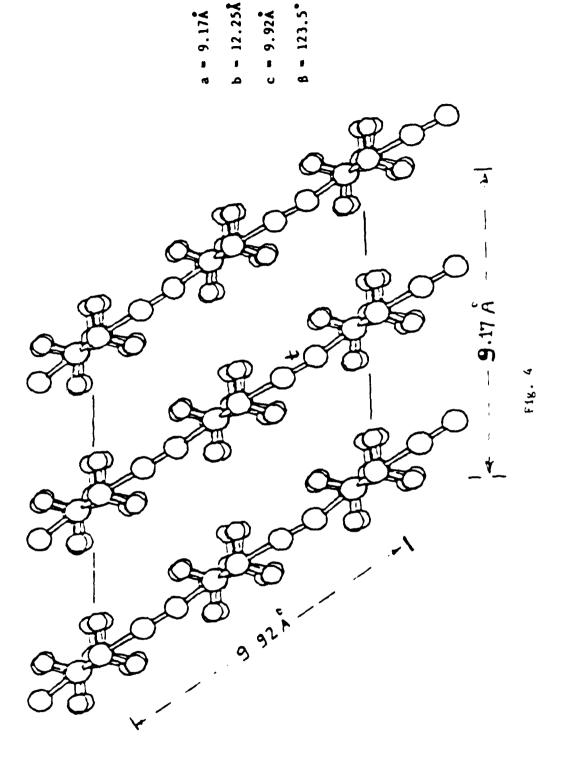
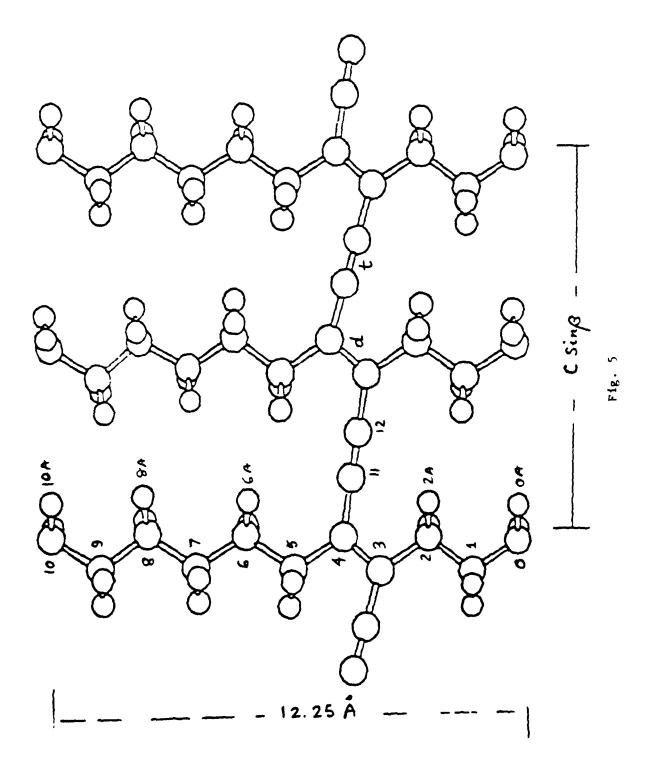


Fig. 3



ac Projection



be Projection

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